# Ethylene route to semiinsulating free-standing GaN

### By using HVPE gallium nitride carbon-doped with ethylene, researchers claim to have achieved record high resistivities.

Researchers from China and Poland have demonstrated that halide vapor phase epitaxy (HVPE) gallium nitride (GaN) carbon-doped with ethylene ( $C_2H_4$  or, showing the double bond,  $CH_2=CH_2$ ) provides record high resistivity for creating semi-insulating freestanding substrates [Qiang Liu et al, Appl. Phys. Lett., v121, p172103, 2022].

The team from Peking University and Sino Nitride Semiconductor Co Ltd in China and Poland's Institute of High Pressure Physics comments: "The resistivity of the sample definitely satisfies the required value  $(10^9\Omega\text{-cm})$  for commercially available semi-insulating GaN substrates. Therefore, such semi-insulating substrates achieved in this work will enable sharp current pinch-off for HEMT devices."

Such substrates allow the implementation of highperformance radio frequency (RF) and microwave power amplifier applications, based on GaN highelectron-mobility transistors (HEMTs). Presently, semi-insulating silicon carbide (SiC) is often used, but the GaN/SiC lattice mismatch introduces increased dislocation densities, impacting performance. A GaN

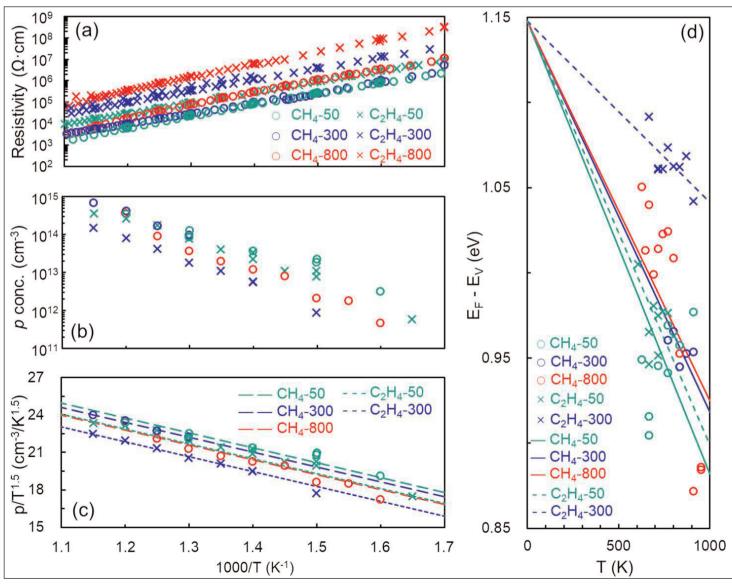


Figure 1. Temperature-dependent resistivity (a) and hole concentration (b) of all samples; (c) fitting curves for hole concentrations shown in (b); and (d) fitting curves to calculate  $E_{AV}(0)$ .

#### Technology focus: Nitride substrates 61

substrate would avoid the lattice mismatch problem. A peak power-added efficiency (PAE) of 82.8% has been reported for GaN HEMTs on free-standing GaN substrate.

The researchers point out that ethylene is an uncommon source for C-doping. However, their work showed that its use results in a 40x higher incorporation of C, relative to the more common methane (CH<sub>4</sub>). This may be due to the ethylene tending to decompose to CH<sub>2</sub>, breaking the double bond, before C incorporation. By contrast, methane would in the first instance dehydrogenate to CH<sub>3</sub> radicals, slowing C incorporation into the GaN growth front.

The substrates were grown on metal-organic vapor phase epitaxy (MOVPE) 4.5µm GaN on sapphire templates (2-inch) and then separated into free-standing material by laser lift-off. The initial HVPE growth consisted of low- and high-temperature layers of GaN (960°C/1070°C) with a V/III ratio of 60. The layer thicknesses were 100µm and 50µm, respectively. The pressure was 900Torr and the source zone of the reactor was kept at 900°C.

Carbon-doped GaN layers were grown on the separated samples with the dopant sources of  $CH_4$  and  $C_2H_4$ diluted by 5% nitrogen. Other factors being approximately equal, the researchers consider that the main difference in C doping from the two sources will arise from their varying chemical and physical properties.

The sample wafer ended up being 750 $\mu$ m thick, which was thinned down to 300 $\mu$ m before dicing into 10mm x 15mm test samples. The thinning was carried out from both sides, removing undoped GaN material, and double-side polishing before dicing.

The average carbon concentration in the samples, determined by secondary-ion-mass spectroscopy (SIMS), increased with dopant gas flow up to 800 standard cubic centimeters/minute (sccm): from  $1.3 \times 10^{17}$ /cm<sup>3</sup> to  $1.7 \times 10^{18}$ /cm<sup>3</sup> for methane, and  $5.8 \times 10^{18}$ /cm<sup>3</sup> to  $1.5 \times 10^{20}$ /cm<sup>3</sup> for ethylene. Silicon contamination from the quartz chamber components was below the level of  $1.3 \times 10^{17}$ /cm<sup>3</sup> at all flow rates.

In terms of input carbon atoms relative to Ga, the team estimates that the incorporation rate of ethylene was 40x higher than for methane.

Hall-effect measurements on 5mm x 5mm testing samples showed p-type conductivity with low mobility less than  $3\text{cm}^2/\text{V-s}$ . Decreases in resistivity at high temperature arose from increased hole concentration (Figure 1). Between  $315^{\circ}\text{C}$  and  $560^{\circ}\text{C}$ , the hole concentration increased from  $10^{12}/\text{cm}^3$  to  $10^{16}/\text{cm}^3$ , while the resistivity fell from  $10^8\Omega$ -cm to  $10^4\Omega$ -cm. The behavior suggests that a single impurity energy level determines the electrical properties of the sample. The team estimates the activation energy at 0K ( $\text{E}_{\text{AV}}(0)$ ) at 1.148eV. This energy corresponds to theoretical expectations for the  $C_N^{0/-}$  acceptor level. The team also found little shift in  $\text{E}_{\text{AV}}$  above 0K, unlike for other reported research by the Institute of High Pressure Physics.

In the highest 800sccm sample doped with ethylene, the p-type conductivity was almost fully self-compensated with very low carrier mobility, making Hall-effect measurements unsuccessful.

The resistivity of GaN:C using 800sccm ethylene at 1000K was  $10^5\Omega$ -cm, a record compared to the previous highest for C-doped GaN of 2000 $\Omega$ -cm or even iron-doped GaN at  $6\times10^4\Omega$ -cm. A problem with iron doping is diffusion to places where it is not wanted. At 833K, the GaN:C-800 had a resistivity of  $6\times10^5\Omega$ -cm, beating a report for manganese-doped GaN of  $2\times10^5\Omega$ -cm.

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