Low threading-dislocationdensity heteroepitaxial AlN film on sapphire

Simple yet effective technique demonstrates potential for mass fabrication of low-cost and high-performance deep ultraviolet devices

In a collaboration that includes Guangzhou University and Peking University, China's Guangdong Institute of Semiconductor Industrial Technology (GISIT) has demonstrated high-quality aluminium nitride (AIN) film grown on sputtered AIN/sapphire. The threading dislocation density (TDD) of the AIN film is reduced to 4.7×10^7 cm⁻², which is reckoned to be an extremely low value for heteroepitaxial AIN film ('High-quality AIN film grown on sputtered AIN/sapphire via growth-mode modification' by Chenguang He, Wei Zhao, Zhitao Chen et al, Crystal Growth & Design, 2018, 18(11) p6816).

Heteroepitaxy of high-quality AIN films is the key to advancing deep-ultraviolet (DUV) devices when a large-size, low-cost AIN substrate is unavailable. To date, however, the prevailing AIN/sapphire template still suffers from a high threading dislocation density. Few reliable methods can obtain high-quality AIN with a TDD of $(3-5)x10^8$ cm⁻². The main causes are the large mismatches in both the lattice constants and the coefficients of thermal expansion between AIN and sapphire, as well as the low surface migration of Al species. The research team proposes a strategy for obtaining high-guality AIN film by combining growth-mode modification with sputtered AIN buffer using metal-organic chemical vapor deposition (MOCVD).

Comparison reveals that the sputtered AIN (d) A buffer consists of smaller and more uniform grains with better c-axis orientation, leading to better growth-mode modification in the subsequent growth process. On one hand, the better c-axis orientation is inherited by the upper AIN epilayer, resulting in a lower screw dislocation density across the whole growth process. On the other hand, the better growth-mode modification significantly suppresses edge dislocations in the upper AIN epilayer by producing high-density nanoscale voids and many 90° bent dislocations.

The formation of nanoscale voids originates from the high-speed lateral growth and large depth/width ratio between three-dimensional islands. Similar to the case of epitaxial lateral overgrowth (ELOG), the pre-existing



Figure 1. Surface morphologies of (a) MOCVD AIN buffer and (b) sputtered AIN buffer taken with scanning electron microscope (SEM) using magnification of 300,000. Crosssectional high-resolution transmission electron microscope (HRTEM) images of (c) AIN film with MOCVD AIN buffer and (d) AIN film with sputtered AIN buffer.

> dislocations can bend and terminate at the local free surfaces provided by these voids, following the principle of dislocation line energy minimization. Because of a higher island density, the AlN film with sputtered AlN buffer has a higher void density of $1.7 \times 10^{10} \text{cm}^{-2}$ in the (0001) plane, providing a higher probability for dislocation termination. During the coalescence processes of the voids, the misorientations between the adjacent domains in the AlN film with sputtered AlN buffer are much smaller, so dislocations at the coalescence boundaries are also effectively suppressed.

The 90° dislocation bending in the areas away from the voids is associated with macro-step movement.

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Figure 2. Cross-sectional dislocation distributions of AIN film with sputtered AIN buffer.

Figure 3. Plan-view TEM images of AIN film with sputtered AIN buffer.

When the macro-step advances laterally and meets with the dislocation in the normal surface, the dislocation will bend as the macro-step and surface advance. The large ratio of lateral growth rate to vertical growth rate results in the 90° dislocation bending. The 90° bent dislocations propagate laterally. Therefore, it is easy for them to come into contact with other dislocations and block them by merging or forming half-loops.

Benefiting from this, only a few threading dislocations can reach the top surface of the AIN film with a sputtered AIN buffer. Plan-view transmission electron microscope (TEM) measurement shows that the TDD of the AIN film with the sputtered AIN buffer is dramatically reduced to an extremely low value of 4.7×10^7 cm⁻², which is 81.2% less than the TDD of the AIN film with an MOCVD AIN buffer.

Additionally, the 5.6 μ m-thick AlN film is crack-free and pit-free. The root mean square (RMS) roughness of the AlN film with sputtered AlN buffer is as small as 0.14nm over 3 μ m x 3 μ m. The concentrations of carbon, hydrogen, and oxygen are as low as 1.6x10¹⁷cm⁻³, 3.0x10¹⁷cm⁻³ and 1.3x10¹⁷cm⁻³, respectively.

The researchers say that the very simple yet effective technique demonstrates potential for the mass fabrication of low-cost and high-performance DUV devices. **https://pubs.acs.org/doi/10.1021/acs.cgd.8b01045**

Figure 4. AIN film with sputtered AIN buffer demonstrates (a) crack-free and (b) atomically flat surface. (c) SIMS profiles of carbon, hydrogen and oxygen impurities.